

# Ni<sub>2</sub>C formation at the graphene/Ni(111) interface: first-principles investigation

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# Outline

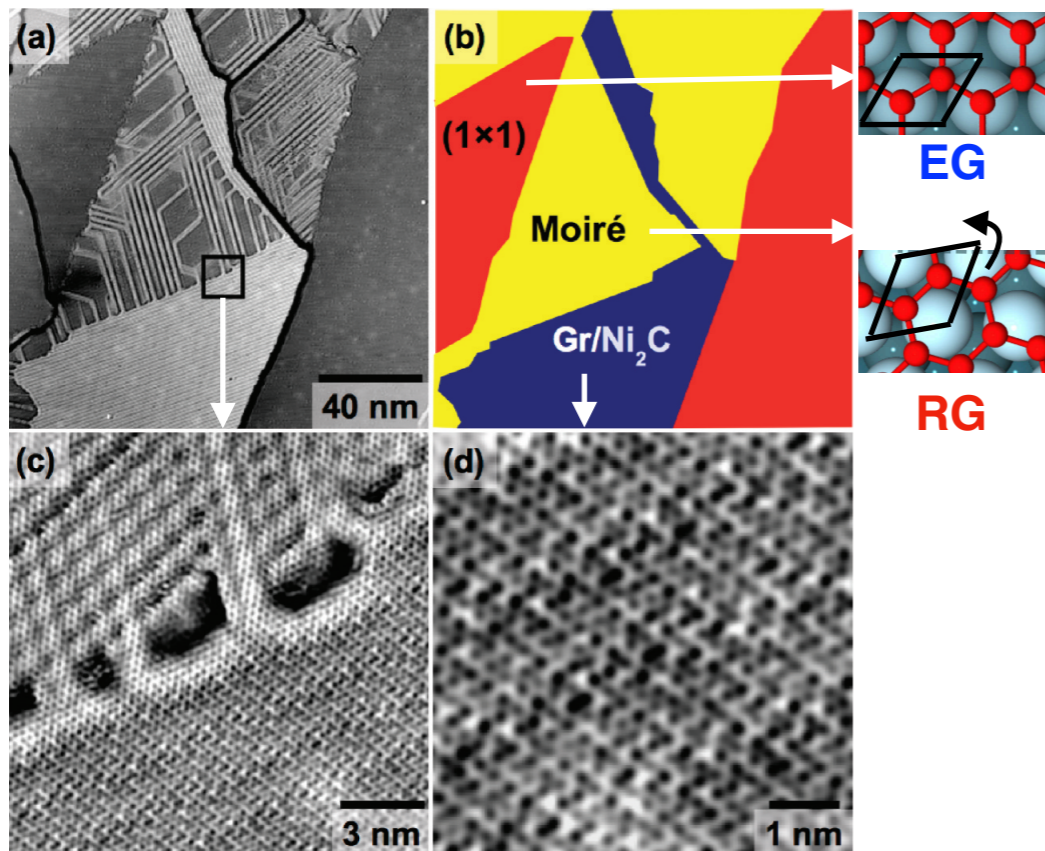
- State-of-the-art
- Simulating and comparing (epitaxial & rotated)-graphene/carbide/Ni(111)
  - Structural models & thermodynamics
  - Electronic properties
- From graphene/Ni interface towards carbide formation

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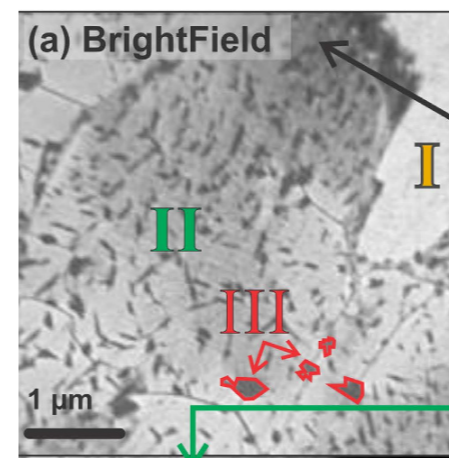
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# State-of-the-art: Experimental facts

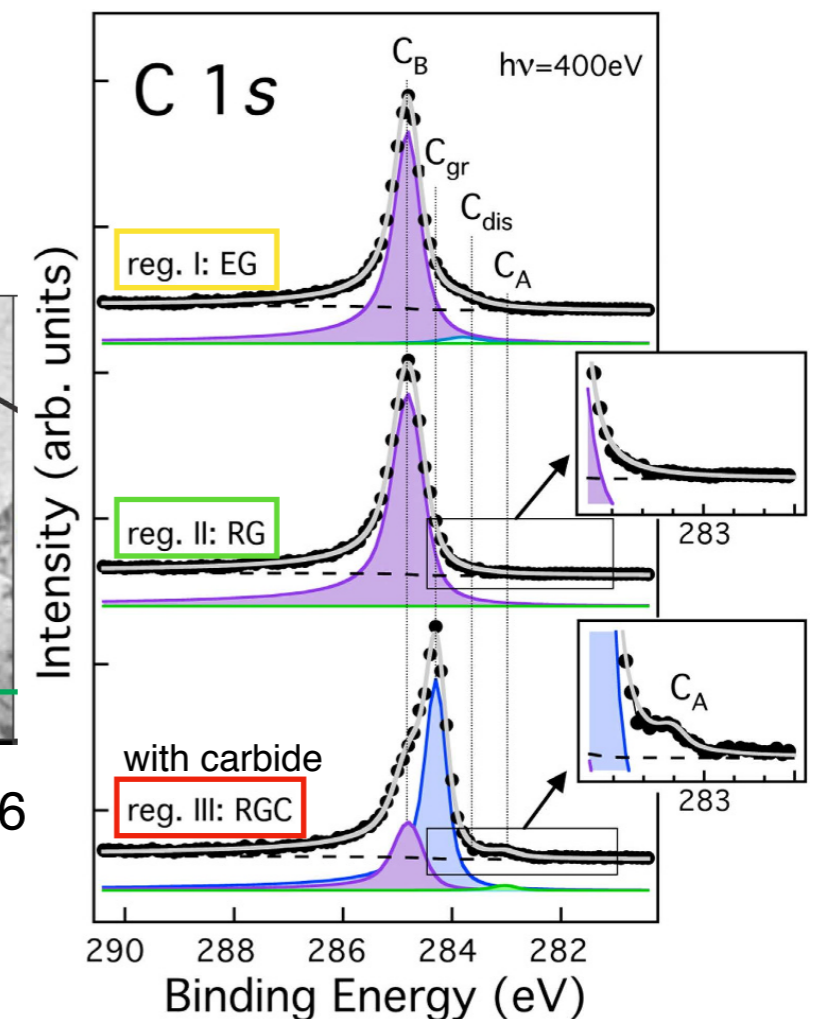
- 1) Coexistence of **EPITAXIAL Graphene (EG)** and **ROTATED Graphene (RG)** on Ni(111)
- 2) **Carbide (Ni<sub>2</sub>C)** can form only **under RG (not EG !)** through surface segregation of C atoms dissolved in Ni bulk, and temperature controls its formation/dissolution



Jacobson et al., ACSNano 2012



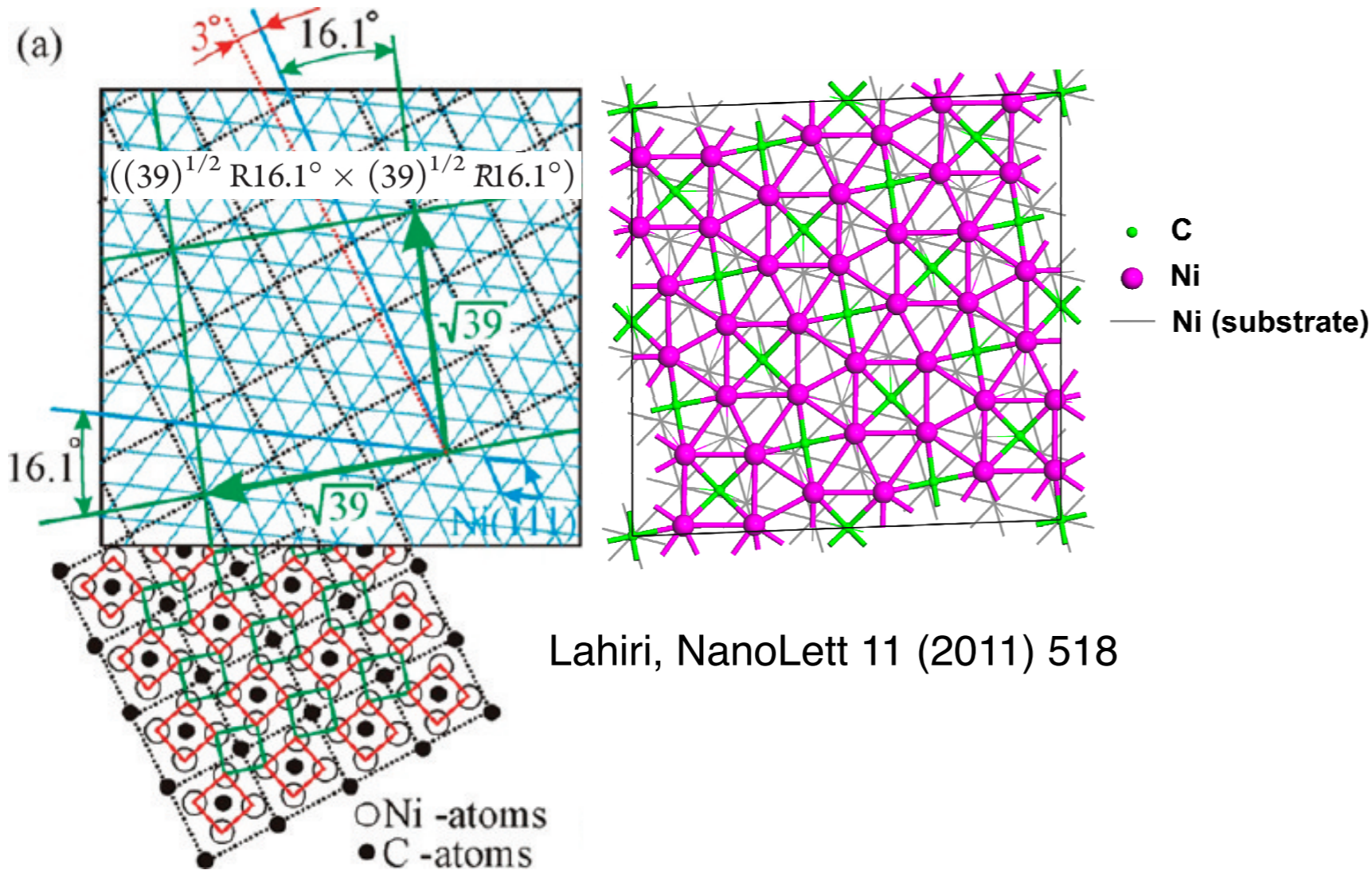
Africh et al., SciRep 2016



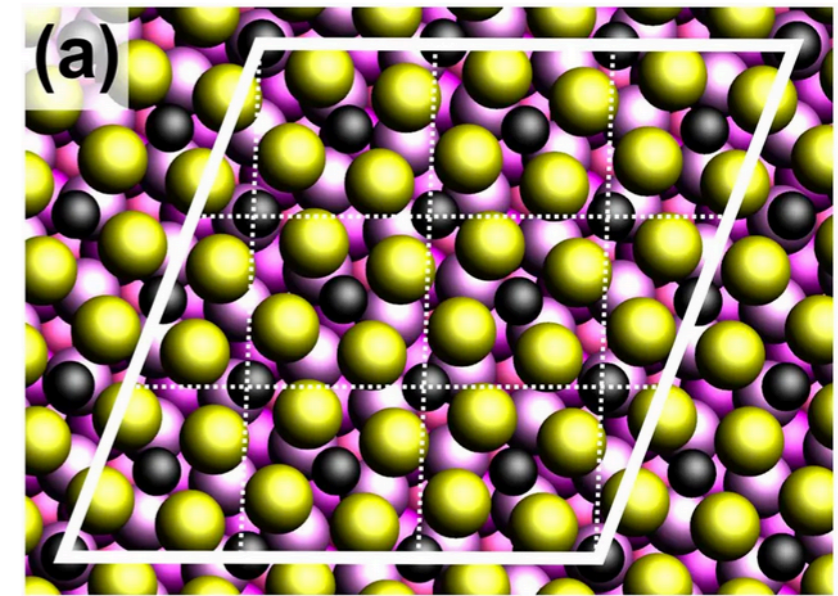
**Why carbide form only under RG?**

**=> investigate and compare EG/Ni<sub>2</sub>C/Ni(111) and RG/Ni<sub>2</sub>C/Ni(111) (if existing) by quantum mechanical numerical simulations**

# State-of-the-art: Models for numerical simulations



Lahiri, NanoLett 11 (2011) 518

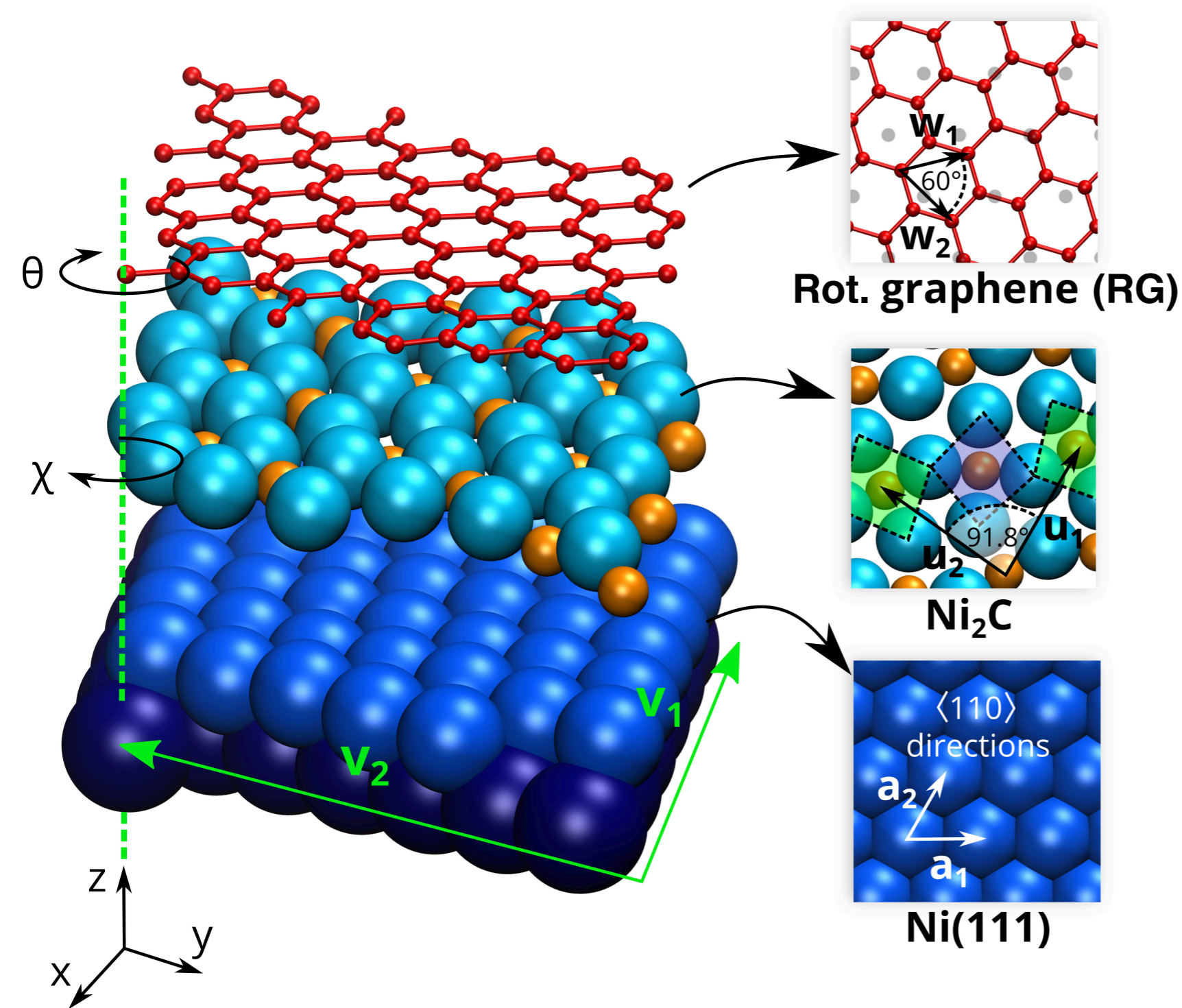


Rameshan et al., Sci Rep. 2018

Models matching  $\text{Ni}_2\text{C}/\text{Ni}(111)$  have been proposed,  
also valid for **EG**/ $\text{Ni}_2\text{C}/\text{Ni}(111)$  (**EG** has a (1x1) matching with Ni(111))

but not valid for **RG**/ $\text{Ni}_2\text{C}/\text{Ni}(111)$

# RG/Ni<sub>2</sub>C/Ni(111): Matching three lattices



A big challenge:  
a coincidence  
moiré cell  
matching **three**  
different lattices



A large cell to be  
investigated using  
ab-initio density  
functional theory  
(DFT)

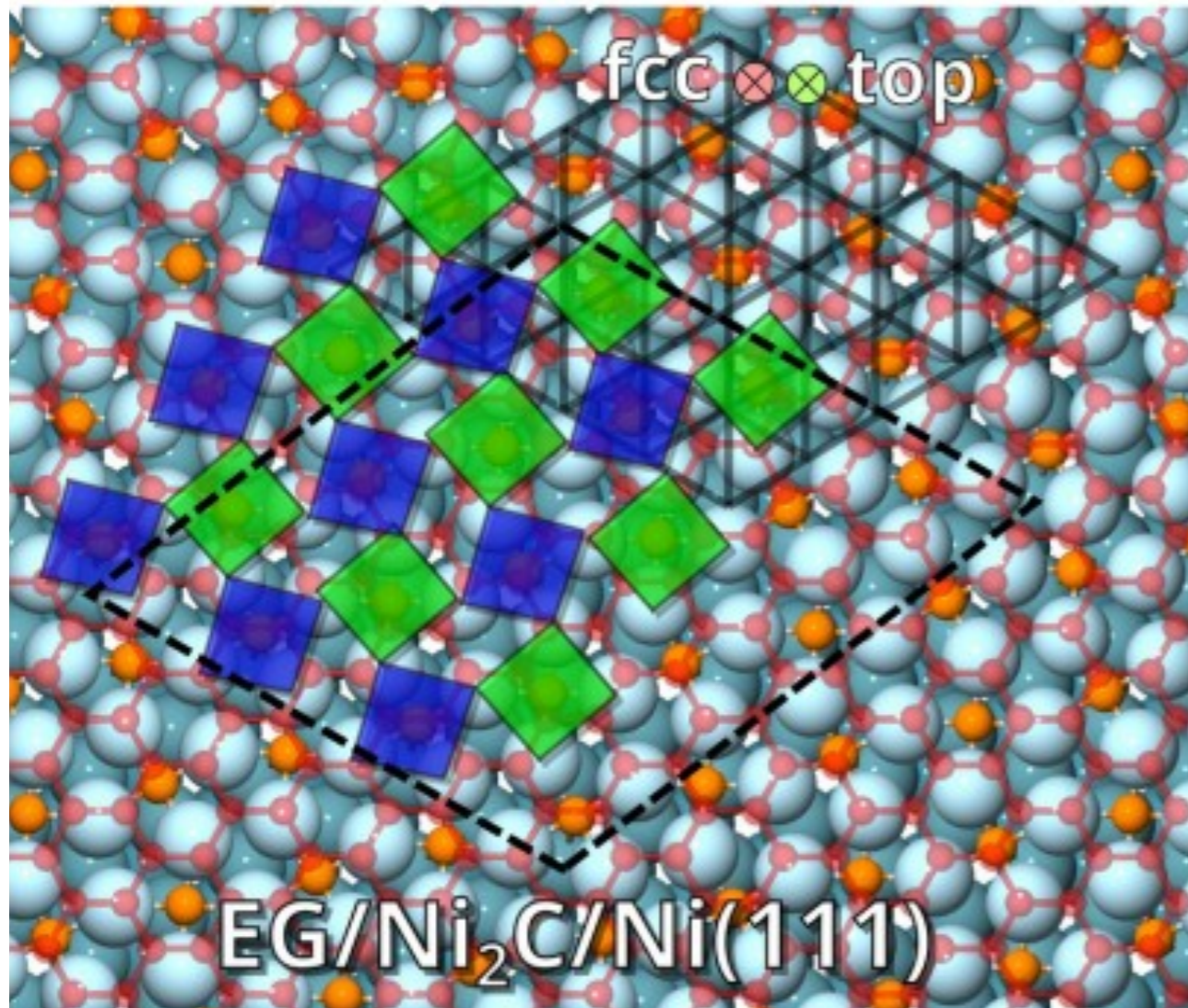


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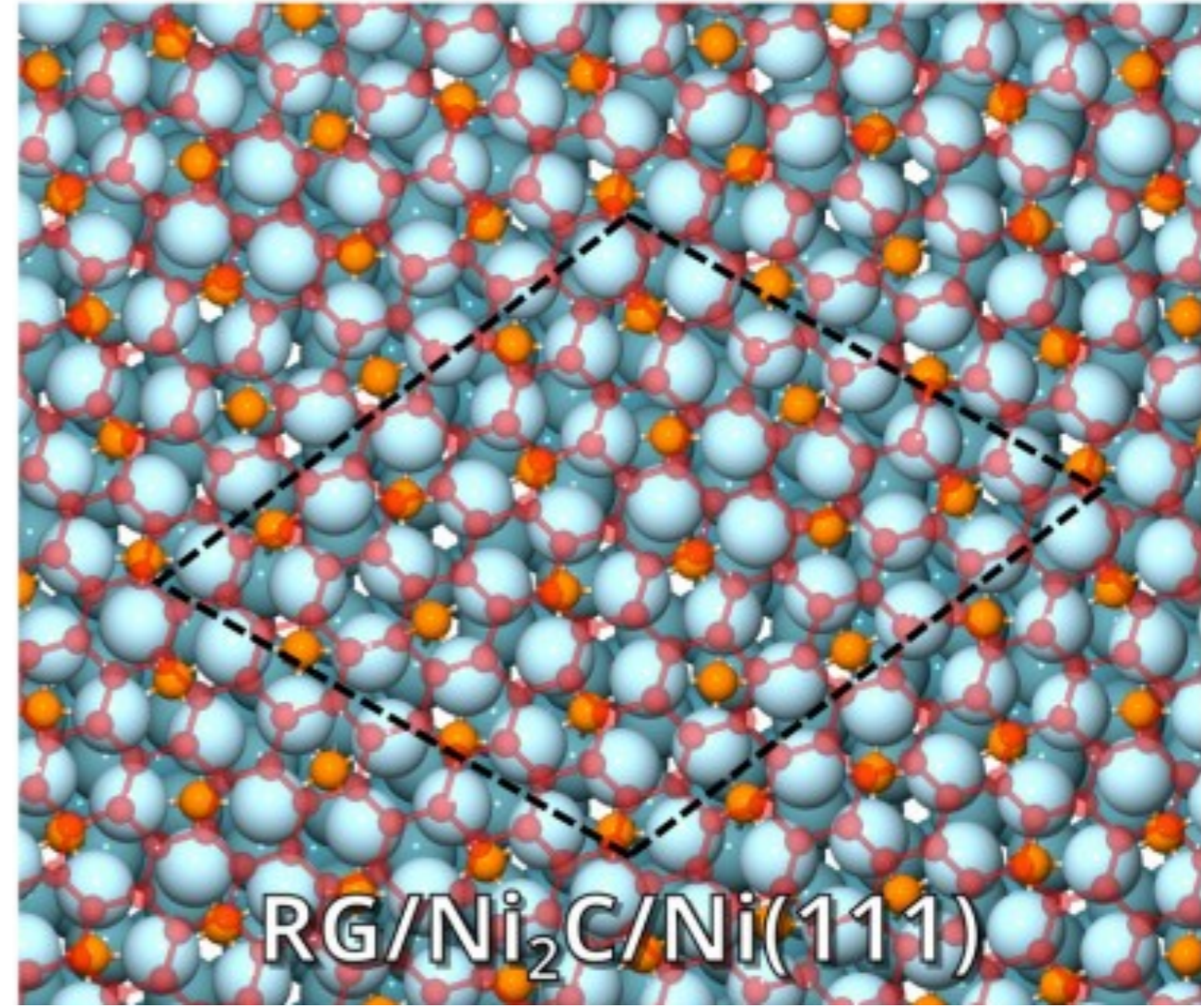
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(epitaxial & rotated)-graphene/carbide/Ni(111)**
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# RG/Ni<sub>2</sub>C/Ni(111) and EG/Ni<sub>2</sub>C/Ni(111): Structural models

cell  $6 \times (\sqrt{43} R7.6)$  accommodating all the three different lattices (RG rotated by  $\sim 17^\circ$ )



**EG** (top-fcc registry w.r.t. Ni) — 222 atoms:  
 Ni(111): 42 Ni / layer (2 layers)  
 Ni<sub>2</sub>C: 18 C + 36 Ni  
 EG: 84 C



**RG** — 226 atoms:  
 Ni(111): 42 Ni / layer (2 layers)  
 Ni<sub>2</sub>C: 18 C + 36 Ni  
 RG: 88 C (slightly compressed)



# RG/Ni<sub>2</sub>C/Ni(111) and EG/Ni<sub>2</sub>C/Ni(111): Structural properties & thermodynamics

From DFT simulations:

- Ni<sub>2</sub>C is thermodynamically stable under both EG and RG
- Ni<sub>2</sub>C detaches both EG and RG from the substrate:

$$h(\text{Gr-Ni}_2\text{C}) = 3.08 \text{ \AA} \quad ( h(\text{Gr-Ni}(111)) = 2.10 \text{ \AA} )$$

$$E_{\text{ads}} = -0.10 \text{ eV/C}_{\text{gr}}$$

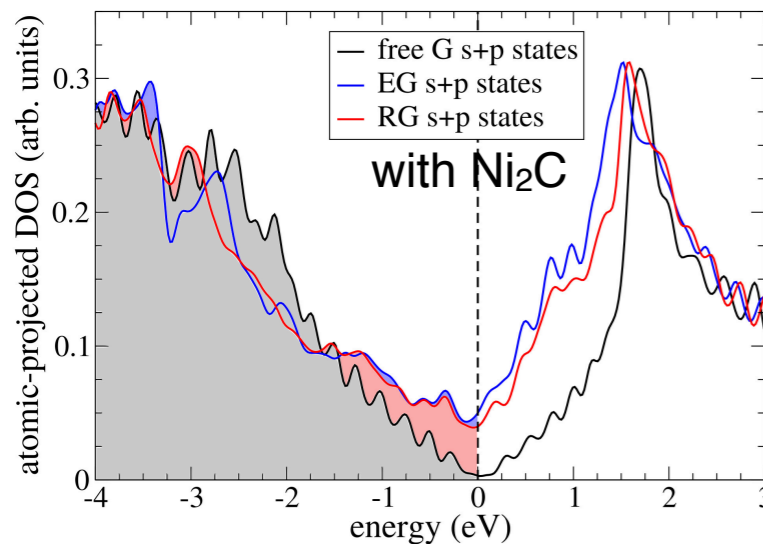
**=> Open question:**

**if Ni<sub>2</sub>C is stable under both EG and RG, what makes the difference?**

# RG/Ni<sub>2</sub>C/Ni(111) and EG/Ni<sub>2</sub>C/Ni(111): Electronic properties

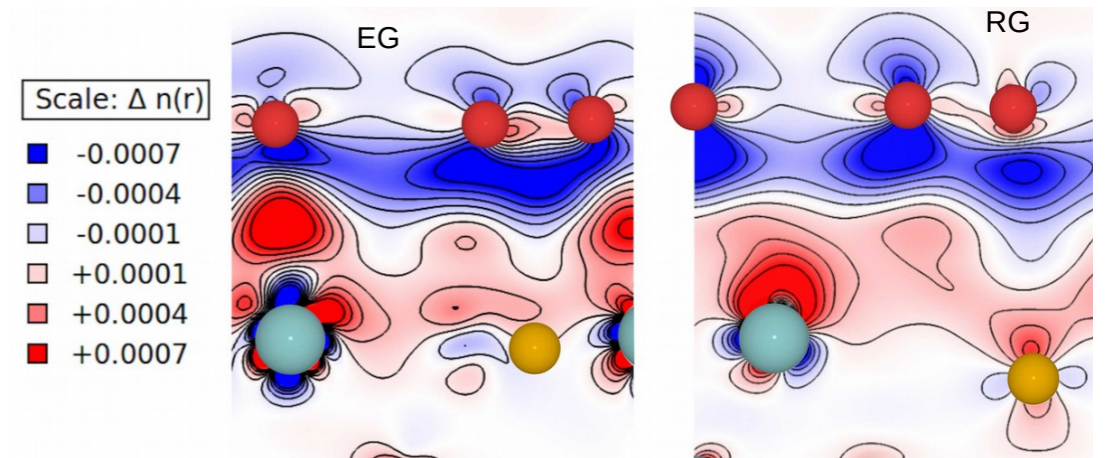
Also very similar for **EG** and **RG**:

- **Graphene DOS** suggests a weak interaction with the substrate

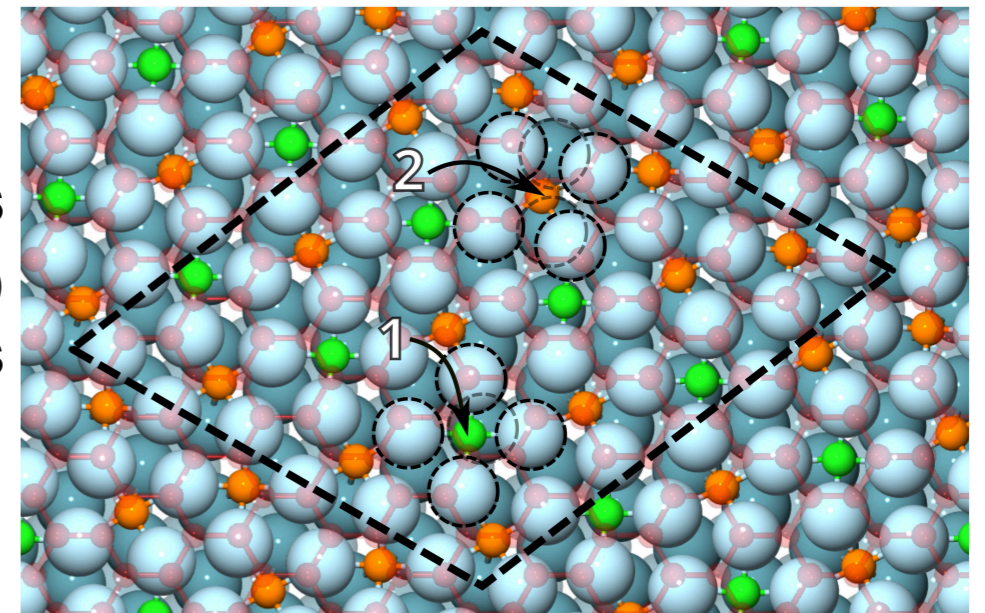
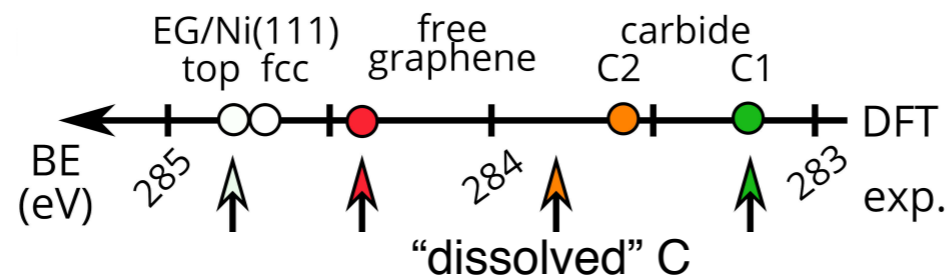


similar to free G (Dirac cone restored); consistent with exp. observation of switching from metallic to semi-metallic behaviour

- small **electron transfer** from G to substrate (differential density plots: whole system - constituents)



- **Calculated C1s binding energies** (core level shifts: fingerprints of different C configurations) allow to interpret the experimental XPS peaks

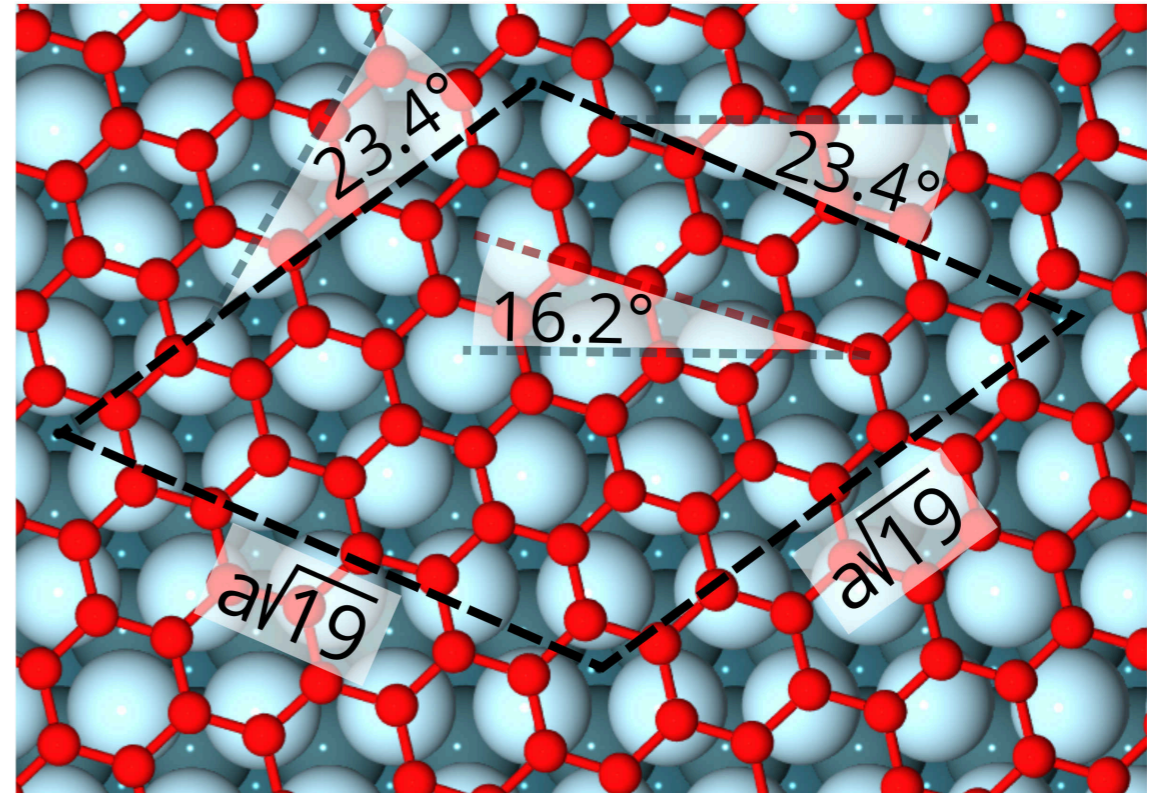
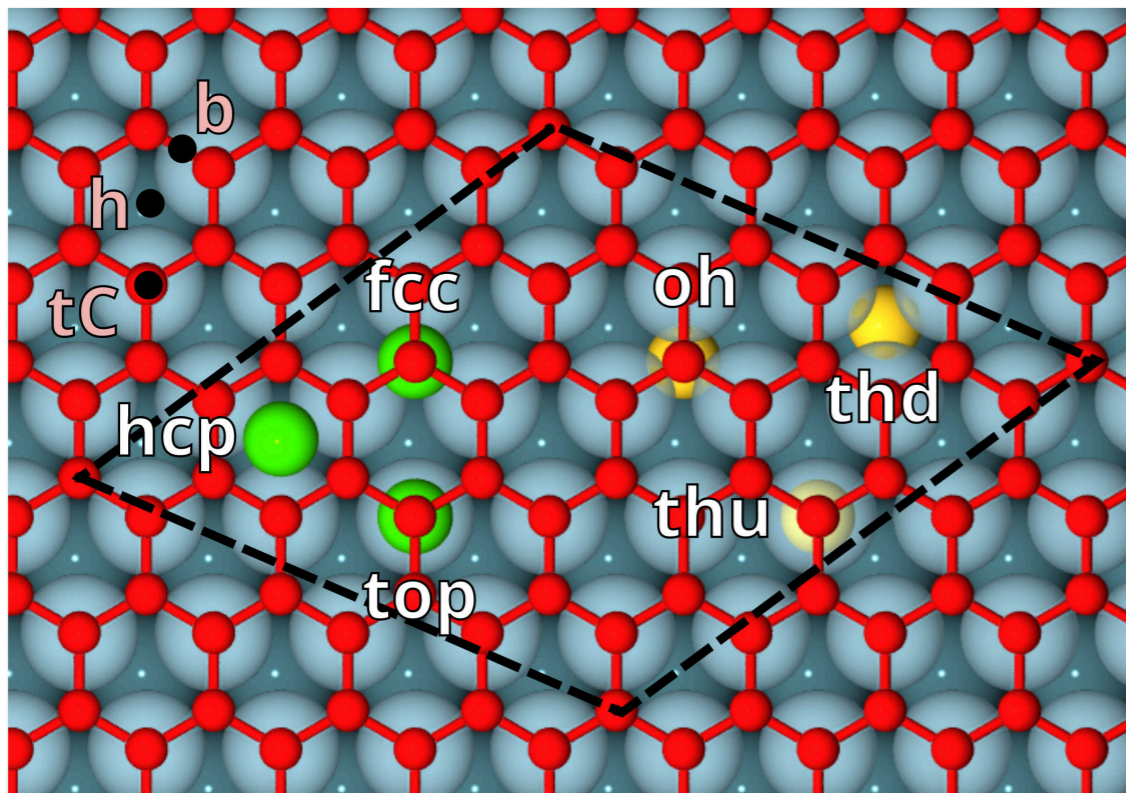


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# RG/Ni(111) and EG/Ni(111): Structural models and properties

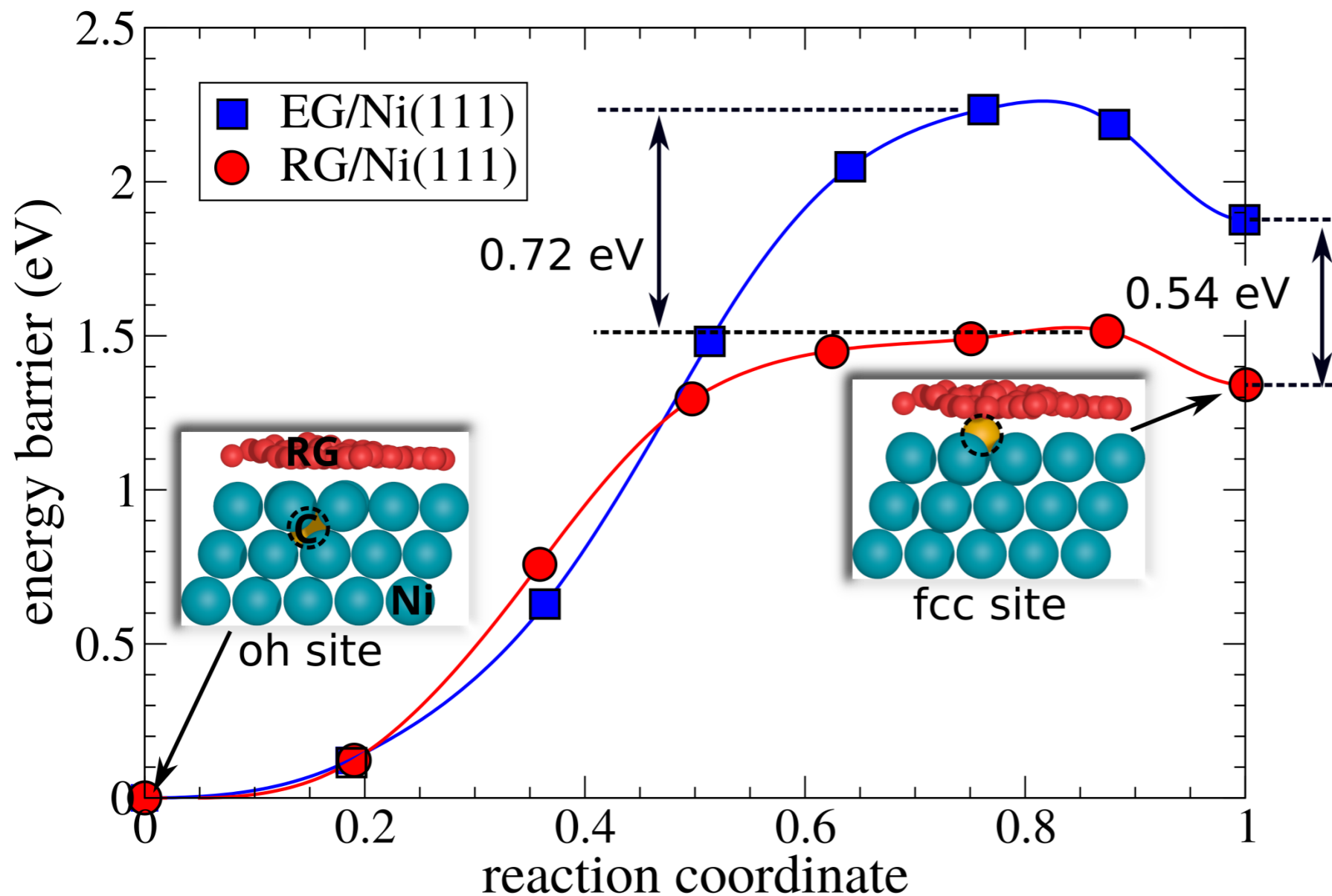
Smallest coincidence cell for RG/Ni(111) (RG rotated by  $\sim 17^\circ$ )  
 $(\sqrt{19} \times \sqrt{19}) R 23.4^\circ$



Starting from these structures, investigate the segregation of C atoms dissolved in Ni bulk towards different surface/subsurface positions:

- oh : the most stable subsurface
- fcc: the most stable on surface

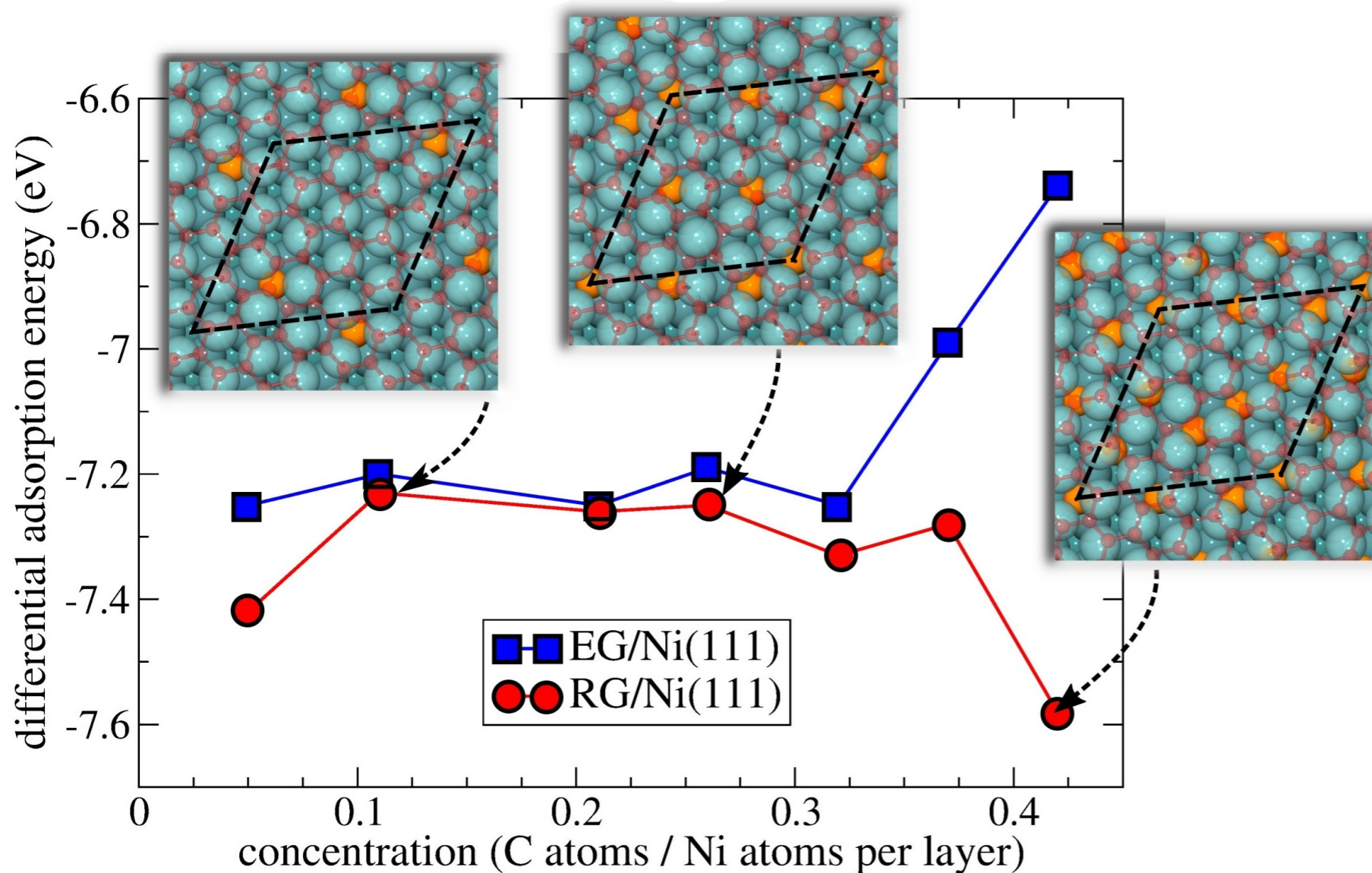
# Towards Ni<sub>2</sub>C formation under RG/Ni(111) and EG/Ni(111)



## Segregation of one individual C atom at a time:

- subsurface *oh* sites are the most stable for dissolved C in Ni both under **RG** and **EG** (compatible with Ni<sub>2</sub>C formation !)
- for small dissolved C concentration, surface segregation is an activated process with high energy barriers (higher for **EG**, but this **does not make a big difference**)

# Towards $\text{Ni}_2\text{C}$ formation under RG/Ni(111) and EG/Ni(111)



## Segregation of further C atoms:

- C concentration < 0.35 ML: similar chemical potential in **RG** and **EG**
- C concentration > 0.35 ML (towards the characteristic C concentration of 0.5 ML in  $\text{Ni}_2\text{C}$ ): segregation much easier under **RG** than **EG** - this **does make the difference!**

# Conclusions

- From thermodynamics, Ni<sub>2</sub>C should be stable at the interface graphene/Ni(111) both under **EPITAXIAL** and **ROTATED** G domains, but it is observed only under **RG**
- Calculated structural and electronic properties of graphene/Ni<sub>2</sub>C/Ni(111) are also very similar for both **EG** and **RG** domains (G detached by Ni<sub>2</sub>C from the surface, recovering ~free-standing features)
- Kinetics arguments for surface segregation of individual C atoms dissolved in Ni bulk do not explain the difference between **EG** and **RG** domains for blocking/favouring Ni<sub>2</sub>C formation
- **Substantial differences between the chemical potentials for high concentration of C atoms segregated on Ni surface under **EG** and **RG** explain the observed Ni<sub>2</sub>C formation only under **RG****

*Thank you for your attention*

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